Dielectric function of α -Sn and its temperature dependence

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The real and imaginary parts of the dielectric function of α -Sn in the (1.2–5.6)-eV photon-energy region have been measured at temperatures between 100 and 350 K. Numerically obtained second derivatives of these spectra show distinct structures attributed to E_1 , $E_1 + \Delta_1$, E'_0 , $E'_0 + \Delta'_0$, E_2 , E'_1 , $E'_1 + \Delta'_1$, and E_3 interband critical points. The line shapes of these structures have been fitted to standard theoretical expressions for various types of critical points. The temperature dependence of the critical-point parameters so obtained is presented. The results are compared with other available data and with theoretical calculations based on the band structure of α -Sn. Data are also presented for the β -Sn obtained after heating α -Sn to temperatures above 70 °C.

I. INTRODUCTION

As shown in the preceding paper, ellipsometry is an excellent method for obtaining the dielectric function of semiconductors versus temperature in the visible, near-ir, and near-uv regions. The quality of the data allows several differentiations as a function of photon frequency. From the so-obtained derivative spectra, interband critical points appear clearly. They can be fitted to standard theoretical line shapes. From these fits the various critical-point (CP) parameters (strengths, Lorentzian broadenings, critical energies, and phase shifts) and their temperature dependence can be accurately obtained.

Gray tin $(\alpha$ -Sn) is a zero-gap semiconductor with diamond structure whose band structure is similar to that of InSb (see Fig. 1 of Ref. 1).3,4 The basic differences between the two materials concern the lack of inversion symmetry in InSb (zinc-blende structure) and the zero gap of gray tin due to the reversal of the order of the 5s antibonding states and the 5p bonding states at the Γ point of the Brillouin zone (BZ), which was discovered by Groves and Paul.⁵ Experimental work on this interesting material has been hampered by the difficulties in preparing single crystals, due to its transformation into metallic β -Sn at a temperature of 13°C (in the bulk).6 Early experiments were performed on lumpy polycrystalline samples obtained by transforming β -Sn at temperatures below 13°C.^{7,8} The growth of single crystals by Ewald and Tufte from mercury solution9 made optical measurements such as ir reflectivity by reflection techniques on growth surfaces possible. Lindquist and Ewald8 obtained, from irreflection measurements, a gap of 0.085 eV, which is now believed to be due to transitions from the $\Gamma_{25'}$ valence band to the L_1 conduction-band minimum.^{3,4} Cardona and Greenaway 10 found, by means of reflectivity measurements in the (1-6)-eV region, the energies of the E_1 , $E_1 + \Delta_1$, E'_0 , E_2 , and E'_1 interband critical points. The measured E_1 and $E_1 + \Delta_1$ energies (1.3 and 1.8 eV) agreed with predictions by Phillips. 11 Other optical measurements on samples grown from the solution followed (reflectivity at oblique incidence, ^{12,13} electroreflectance, ^{12,14} resonant Raman scattering ¹⁵). These results showed some

discrepancies with each other and with theoretical calculations of the dielectric function, 3,4,13,16 especially concerning the height of the E_2 peak in the reflection spectrum. The influence of the poor quality of the crystals on these measurements, with Hg occlusions and an unknown amount of dissolved Hg, remained an open question. 17 Agreement between measured and calculated 3,4,10-14,16,18-25 critical-point energies is, however, generally good.

A new method of growing high-quality α -Sn films has been reported recently:²⁶ films of α -Sn up to a thickness of 0.5 μ m were prepared with the molecular-beam-epitaxy technique at room temperature (RT). These films showed a much higher transformation temperature (about 70°C) than the crystals grown from Hg solution. This temperature depends on the surface orientation and the film thickness, decreasing with increasing thickness.²⁷ These films are much flatter and have larger areas than the surfaces of samples grown from the mercury solution.9 Thus they are particularly suitable for ellipsometric measurements. They also have been used to determine the dispersion relations of the valence bands by means of angle-resolved photoemission.²⁸

In this paper we report ellipsometric measurements of the dielectric function of α -Sn prepared on InSb by the molecular-beam-epitaxy technique.26 The measurements were performed at several temperatures between 100 and 350 K. Special emphasis was placed on the investigation of the parameters of the observed critical points as a function of temperature. Data were also obtained for films transformed into metallic β -Sn after heating to 90 °C.

In the following section we will give a short description of the sample preparation and experimental arrangement of the ellipsometer. In Sec. III we present the experimental data. A discussion, comparison with band-structure calculations, and the results of earlier experiments are given in Sec. IV.

II. EXPERIMENTAL DETAILS

The α -Sn sample was a 950-Å-thick film grown on a 7×7-mm² (001)-oriented InSb surface using the

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molecular-beam-epitaxy technique of Farrow et al. ²⁶ The growth, at a typical rate of 2 Å s^{-1} and monitored by reflection high-energy electron diffraction (RHEED), took place under ultrahigh-vacuum conditions $(3 \times 10^{-10} \text{ Torr})$. The cleaness of the film was monitored by photoemission spectroscopy, ²⁸ and there was no indication of surface oxidation. Further details of sample preparation can be found in Refs. 28 and 29. To avoid surface contamination, the α -Sn sample was transferred inside a glove bag in an Ar atmosphere into an ultrahigh-vacuum Dewar. The sample was mounted on the cold finger of a liquid-nitrogen cryostat. The temperature was regulated by means of a resistance heater inside a copper holder. Temperature stability was typically $\pm 1 \text{ K}$.

Measurements were taken with an automatic rotating-analyzer ellipsometer. A tungsten quartz-iodine lamp was used in the (1.2-2.3)-eV range; a 75-W short-arc Xe lamp was used for photon energies above 1.8 eV. The light was dispersed by a $\frac{3}{4}$ -m Spex single monochromator, and polarized and analyzed with Rochon-quartz prisms. Photomultipliers with S20 and S1 responses were used in combination with the tungsten and Xe lamps, respectively. All measurements were performed at an angle of incidence of 67.5°. Further details concerning the experimental arrangement can be found in Ref. 30.

III. RESULTS

The complex dielectric function (ϵ) was obtained from the measured ellipsometric angles $(\Psi \text{ and } \Delta)$. The high-vacuum conditions of sample preparation and measurement allow us to use a simple two-phase model³¹ to evaluate ϵ from the complex reflectance ratio. Figure 1 shows the real (ϵ_1) and imaginary (ϵ_2) parts of ϵ between 1.2 and 5.6 eV at two selected temperatures. Data collected with both multipliers are matched smoothly in the region of overlap. A 10-meV mesh was used for all temperatures. The structures E_1 , E_0' , E_2 , E_1' , and E_3 shift to lower energies and broaden with increasing T. To obtain the

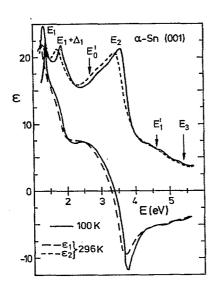


FIG. 1. Real (ϵ_1) and imaginary (ϵ_2) parts of the dielectric function of an α -Sn(001) film on InSb at 100 and 296 K.

critical-point parameters and resolve the fine structure, we calculate numerically³² the second-derivative spectra $(d^2\epsilon/d\omega^2)$ of the complex dielectric function from our ellipsometric data. The experimental $d^2\epsilon_1/d\omega^2$ (dots) is displayed in Fig. 2, together with the corresponding fits to critical-point line shapes for the real and imaginary parts of the different gaps. Only structure that appeared consistently at different T's is labeled in this figure. The vertical scale in the regions between $E_1 + \Delta_1$ and E_2 and above the E_2 CP is enlarged by a factor of 20. A leastsquares procedure, with both $d^2\epsilon_1/d\omega^2$ and $d^2\epsilon_2/d\omega^2$ fitted simultaneously, was used for fits. Two-dimensional (2D) CP line shapes were employed for the E_1 , $E_1 + \Delta_1$, E_2 , E_1' , and $E_1' + \Delta_1'$ singularities. Excitonic effects were taken into account, allowing for a mixture of two CP's;33 the mixture of a 2D minimum and a saddle point can be written as34,35

$$\epsilon \sim C - \ln(E - \omega - i\Gamma)e^{i\phi}$$
, (1)

where E is the CP energy, ω is the photon energy, Γ is the broadening parameter, and ϕ is a phase angle giving the amount of mixture ($\phi=0$ represents a pure minimum and $\phi=\pi/2$ represents a saddle point).

In the case of 3D CP's, the dielectric function can be approximated by^{34,35}

$$\epsilon \sim C + i(\omega - E + i\Gamma)^{1/2} e^{i\phi}$$
 (2)

An M_0 CP corresponds to $\phi=0$, an M_1 CP to $\phi=\pi/2$. Equation (2) was used to fit the E_0' , $E_0'+\Delta_0'$, $\Delta_7^v-\Delta_6^v$, and E_3 transitions.

The E_0' and $E_0' + \Delta_0'$ structure were fitted simultaneously: the spin-orbit (SO) splitting (Δ_0') of 0.3 eV obtained at 100 K was kept fixed in the fit at higher temperatures. The same procedure was used for E_1' and $E_1' + \Delta_1'$ with $\Delta_1' = 228$ meV. The $\Delta_7^v - \Delta_6^c$ structure was fitted, together with $E_0' + \Delta_0'$, keeping the parameters obtained previously for $E_0' + \Delta_0'$ fixed.

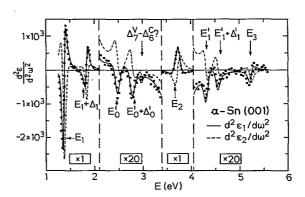
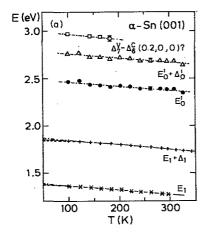


FIG. 2. Experimental second derivatives with respect to the photon energy of the real part of the dielectric function $(d^2\epsilon_1/d\omega^2)$ of α -Sn at 100 K (dots). Lines correspond to the best fits to the real and imaginary parts with different types of critical points, as explained in the text. The regions corresponding to the E_0' transitions and above the E_2 CP are enlarged by a factor of 20. Because of its uncertainty, we have put a question mark on the $\Delta_0^p - \Delta_0^c$ assignment of the structure at 3.0 eV.

TABLE I. Values of the parameters obtained by fitting the critical-point energy (E) vs temperature (T) to the equations $E(T)=E(0)-\alpha T^2/(\beta+T)$, $E(T)=E_B-\alpha_B[1+2/(e^{\Theta/T}-1)]$, and $E(T)=E_L-\gamma T$. Values in parenthesis indicate 95% reliability of the fits. The assignment of the $\Delta_T^p-\Delta_0^c(?)$ structure is uncertain.

	E(0) (eV)	$(10^{-4} \text{ eV K}^{-1})$	β (K)	E_B (eV)	$lpha_B$ (eV)	Θ (K .)	$egin{aligned} E_L \ ext{(eV)} \end{aligned}$	$(10^{-4}\mathrm{eV}\mathrm{K}^{-1})$
$\overline{E_1}$	1.38(1)	6(2)	230(100)	1,44(4)	0.09(4)	340(90)	1.405(6)	4.6(3)
								5.4(6) ^a
$E_1 + \Delta_1$	1.853(5)	7(4)	270(140)	1.97(4)	0.13(4)	390(80)	1.894(7)	4.9(3)
								4.2(6) ^a
E_0'							2.51(2)	4.5(9)
								1.5(5) ^a
$E_0' + \Delta_0'$							2.81(2)	4.5(9)
								$\pm 2^{a}$
$\Delta_7^v - \Delta_6^c(?)$							3.02(2)	5.(1)
E_2	3.748(8)	7(2)	220(130)	3.82(3)	0.08(3)	310(90)	3.780(5)	5.1(2)
					•			3.5(6) ^a
E_1'							4.39(4)	6.(2)
$E_1' + \Delta_1'$		•					4.62(4)	6.(2)
E_3			•				5.3(2)	8.(3)

aReference 14.



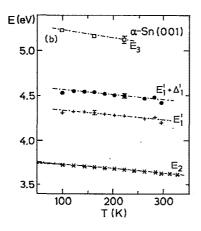


FIG. 3. Dependence on temperature of the critical-point energies of α -Sn. Solid lines represent the best fit of the data to Eq. (3), dashed lines, to Eq. (4), and dotted-dashed lines, to Eq. (5).

The critical energies of the analyzed structures are plotted in Figs. 3(a) and 3(b) versus T. In Table I we have listed the fit parameters of Varshni's empirical formula, ³⁶

$$E(T) = E(0) - \frac{\alpha T^2}{T + \beta}, \qquad (3)$$

and those for an expression proportional to Bose-Einstein statistical factors, 30

$$E(T) = E_B - \alpha_B \left[1 + \frac{2}{e^{\Theta/T} - 1} \right], \tag{4}$$

as well as the coefficients of a linear fit,

$$E(T) = E_L - \gamma T . \tag{5}$$

The uncertainties in parenthesis in Table I represent 95% reliability.

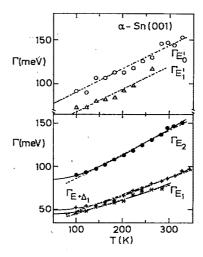


FIG. 4. Dependence on temperature of the critical-point broadening parameters of α -Sn. Solid lines represent the best fit to an expression similar to Eq. (4), and dashed lines represent the best linear fit.

TABLE II. Values of the parameters obtained by fitting the Lorentzian broadenings vs temperature to the equations $\Gamma(T) = \Gamma_1 + \Gamma_0[1 + 2/(e^{\Theta/T} - 1)]$ and $\Gamma = \Gamma_L + \gamma T$. The temperature Θ was forced to be the same as in Table I. Values in parenthesis represent 95% reliability of the fits.

	Γ_L (meV)	γ (10 ⁻⁴ eV K ⁻¹)	$\Gamma_{\rm i}$ (meV)	Γ_0 (meV)	© (K.)
$\overline{E_1}$	28(4)	1.7(2)	5(7)	40(5)	350
$E_1 + \Delta_1$	28(4)	2.0(2)	-2(3)	51(2)	390
E_0'	64(9)	2.7(4)			
E_2	58(4)	2.7(2)	31(3)	53(2)	310
E_1'	42(14)	2.6(8)			

The Lorentzian broadening parameter Γ for the E_1 , $E_1 + \Delta_1$, E_2 , E_1' , and E_0' CP's are displayed in Fig. 4 as a function of T. The coefficients of the fits to an expression similar to Eq. (4) and a linear expression are represented in Table II. The broadenings of the $E_0' + \Delta_0'$ and $E_1' + \Delta_1'$ CP's are, within the experimental error, equal to those of their spin-orbit-split partners. The error bars for $\Gamma(E_3)$ and $\Gamma(\Delta_0^2 - \Delta_0^2)$ are too large to allow us to obtain their temperature dependence.

The phase angles ϕ representing the mixture of contiguous CP's are plotted in Fig. 5 versus T for the $E_1+\Delta_1$ and E_2 transitions; the values of ϕ for the E_2 CP of Ge obtained in recent measurements³⁰ are also displayed for comparison. For $\pi/2 < \phi < \pi$, as found from our fit, Eq. (1) represents a mixture of a 2D saddle point with a maximum.

The optical constants of the white-tin film obtained after the phase transition produced by heating to 90 °C are shown in Fig. 6. They display the Drude behavior typical of metals.

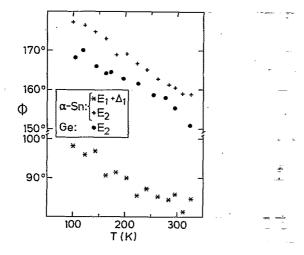


FIG. 5. Dependence on temperature of the excitonic parameter ϕ defined in Eq. (1) for the $E_1 + \Delta_1$ and E_2 CP's of α -Sn and for the E_2 CP of Ge (dots).

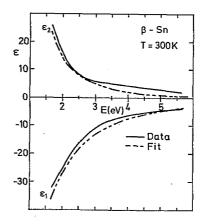


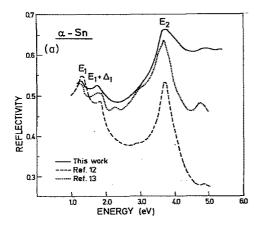
FIG. 6. Pseudodielectric function of the β -Sn film obtained upon transforming α -Sn films by heating to 90 °C measured at RT (solid lines). The dashed lines represent the best fit of the data to Eq. (12).

IV. DISCUSSION

A. General

Some comments about the sample conditions are warranted. The carrier concentrations of similarly grown samples have been shown to be lower than 4×10^{19} cm⁻³.26 Doping below this level has a negligible influence on the transitions studied in this work.³⁷ The presence of microcrystallites or amorphous α -Sn is also ruled out from the measurements of Ref. 27: This would lead, for example, to asymmetric broadening³⁸ of the Raman line, which has not been observed.²⁷ Angle-resolvedphotoemission experiments²⁸ indicate that no β -Sn regions are present in the (001) films below the transition temperature. The lattice constants of α -Sn and InSb are 30 6.4892 and 6.479 37 Å, respectively; as a result of this lattice mismatch, at the interface the α -Sn film is laterally compressed, and therefore an elastic dilation along [001] results. The hydrostatic part of this strain should cause a blue shift of the energy gaps, 40 and the uniaxial one leads to a splitting of the energy levels. The last term is usually small and can be neglected. We have estimated the former for the E_1 and E_2 CP's using the pressure coefficients of these gaps for InSb (Ref. 39) (those of α -Sn are not available): blue shifts of 14 and 9 meV are obtained with respect to bulk unstrained α -Sn, respectively. These differences should remain nearly constant or slightly decrease with increasing T due to the comparable linear thermal-expansion coefficients of α -Sn and $(4.7 \times 10^{-6} \text{ and } 5.37 \times 10^{-6} \text{ K}^{-1}, \text{ respectively}^{39})$

In Figs. 7(a) and 7(b) the normal-incidence reflectivity spectrum obtained from the measured dielectric function at RT $[R=|(\epsilon^{1/2}-1)/(\epsilon^{1/2}+1)|^2]$ is presented and compared with earlier experimental work and theoretical calculations. Although the agreement in the peak positions in Fig. 7(a) is remarkably good, larger differences are seen in the absolute values of the reflectivity. This fact is probably due to different sample characteristics



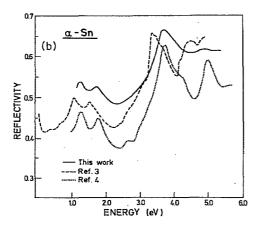


FIG. 7. (a) Normal reflectivity (R) spectrum of α -Sn at RT from the present work and Refs. 12 and 13. Note the differences in the reflectivity values. (b) Comparison of the present data with calculated reflectivity spectra (Refs. 3 and 4).

and mainly to surface conditions. Typically, the magnitude of the E_2 structure and the ratio of the $E_1 + \Delta_1$ to E_1 peaks decrease if oxides or adsorbed layers are present.41 In view of the precautions taken to keep the surface clean, and the good agreement with theory (Fig. 7), we believe that the present measurements yield the most reliable optical constants obtained so far for α -Sn. In Table III we list the CP energies of the main structures in the optical spectra of α -Sn above the fundamental gap taken from the literature, together with the present data. While the best agreement in absolute values of reflectivity with theory is found for the first-principles relativistic orthogonalized-plane-wave calculations of Ref. 3 [Fig. 7(b)], the theoretical critical energies differ in this case by as much as 300 meV from the measured ones. The nonlocal empirical pseudopotential calculations of Ref. 4 obtain CP energies in much better agreement with the present

A notable difference between the optical constants of gray Sn and the other two group-IV semiconductors is the magnitude of the E_2 peak in ϵ_2 compared with that of the E_1 and $E_1 + \Delta_1$ CP's, which is much higher in the case of Si (Ref. 37) and Ge (Ref. 30) than for α -Sn (see Fig. 1). This observation has also been made in Ref. 12, although in this case the E_2 peak is likely to be lowered by surface contamination. We feel that in our case this effect is not due to contamination, but is derived directly from a shift of oscillator strength to lower energies with increasing atomic number. An estimate of the ratio between $\epsilon_2(E_2)$ of Ge and gray tin can be made from the sum rule:⁴²

$$n_{\text{eff}}(\omega_{M}) = \frac{2m}{4\pi^{2}e^{2}N_{\text{at}}} \int_{0}^{\omega_{M}} \omega' \epsilon_{2}(\omega') d\omega'$$
$$= \frac{2m}{4\pi^{2}e^{2}N_{\text{at}}} I(\omega_{M}) , \qquad (6)$$

TABLE III. Critical-point energies (eV) of interband transitions in gray tin. The corresponding temperatures are given in the foot-

E_1	$E_1 + \Delta_1$	E_0'	$E_0' + \Delta_0'$	$\Delta_7^v - \Delta_6^c(?)$	E_2	E_1'	$E_1' + \Delta_1'$	$E_1' + \Delta_1' + \Delta_1$	E_3
				Exper	iment		,	-	
1.28(2)a	1.76(2)a			-	3.66(2) ^a				
1.25 ^b	1.80 ^b	2.25 ^b					_		
1.25°	1.65°	2.25°			3.02°				
1.365(2)d	1.845(4) ^d	2.28 ^d	2.63 ^d		3.718(2) ^d	4.11 ^d	4.39 ^d	4.89 ^d	
1.316(5)°	1.798(6) ^e	2.42(3)e	2.72(3) ^e	2.94(3) ^e	3.681(4)°	4.28(4) ^e	4.51(4)°		5.15(5) ^e
				The	eory				
1.10 ^f	1.55 ^f	1.96 ^f	2.45 ^f	2.7 ^f	3.35 ^f	4.2 ^f	4.4 ^f	4.8 ^f	5.4 ^f
1.378	1.45 ⁸	2.68g	2.72g	2.948					
1.33 ^h	1.99 ^h	2.05 ^h	2.75 ^h		3.77 ^h	4.14 ^h	4.51 ^h	5.17 ^h	
1.34 ⁱ	1.83 ⁱ	2.08 ⁱ	2.66 ⁱ	2.91 ⁱ	3.78 ⁱ	4.68 ⁱ			

^{*}Reference 10 at 200 K.

^bFrom reflectance in Ref. 13 at 278 K.

^cFrom ϵ in Ref. 13 at 278 K.

^dReference 14 at 205 K.

This work at 200 K.

fReference 3.

Reference 13.

^hReference 25.

Reference 4.

where $N_{\rm at}$ is the atom density and m(e) is the electron mass (charge); the integral

$$I(\omega_M) = \int_0^{\omega_M} \omega' \epsilon_2(\omega') d\omega' \tag{7}$$

is inversely proportional to the atomic volume $(V_{\rm at})$. Using in Eqs. (6) and (7) the approximation $I \simeq \omega(E_2)\epsilon_2(E_2)$, a ratio

$$\frac{[\epsilon_2(E_2)]_{Ge}}{[\epsilon_2(E_2)]_{\alpha-Sn}} = \frac{[\omega(E_2)V_{at}]_{\alpha-Sn}}{[\omega(E_2)V_{at}]_{Ge}} = 1.3$$
 (8)

is obtained. The experimental value of 1.4 for this ratio is in very good agreement with that predicted in Eq. (8).

B. E_1 transitions

The first doublet seen in Fig. 1 below 2 eV corresponds to the structures labeled E_1 and $E_1 + \Delta_1$. Band-structure calculations localize the transitions responsible for these structures at the L point,4 or in a region between the point $\vec{k} = (\pi/3a)(1,1,1)$ and the L point,³ from the upper valence $(\Lambda_{4.5})$ to the lowest conduction bands (Λ_6) (E_1) , and from the Λ_6 valence band to the same conduction band $(E_1 + \Delta_1)$. A 3D M_0 critical point at L and an M_1 critical point at $(\pi/24a)(1,1,1)$, with the M_0 CP slightly lower in energy, have also been associated with these transitions.¹³ The observed line shape is compatible with either a 3D M_1 CP or a 2D minimum. The differences between the description of these critical points with either 3D or 2D line shapes has been discussed in detail in Ref. 30. We believe that 2D critical points provide, at present, the most reasonable and internally consistent representation of these transitions.

The second derivatives of the E_1 and $E_1 + \Delta_1$ structures were fitted simultaneously, allowing for excitonic effects with a mixture of a 2D minimum and a saddle point [Eq. (1)]. The best fit of $d^2\epsilon/d\omega^2$, together with the experimental data for the real part of the dielectric function at 100 K, are shown in Fig. 2. The CP energy of the E_1 transition found in the present work (see Table III) is ~ 50 meV higher than the values obtained with reflectance, 10,13 and the same amount lower than the electroreflectance results.14 Taking into account the influence of the InSb substrate (see Sec. IV A) brings the discrepancy to ~ 40 meV with the former results and ~60 meV with the latter results. Line-shape effects are known to be responsible for the discrepancies between peaks in $\Delta R/R$ and in $\Delta \epsilon_1$ and $\Delta \epsilon_2$); ^{30,43} therefore, we believe that our complete lineshape analysis of the CP's should give the most reliable results. Except for Ref. 3, theoretical values of the E_1 and $E_1 + \Delta_1$ energies are only slightly higher (~20 meV) than the present results. The SO splitting $\Delta_1 = 482 \pm 6$ meV found in our study agrees reasonably well with previous experimental values and rather well with theoretical values, with the exception of the self-consistent nonlocal pseudopotential calculations of Ref. 25 (see Table IV). This SO splitting remains constant at all temperatures with an uncertainty no larger than the experimental error $(\pm 6 \text{ meV}).$

The broadening parameters of E_1 and $E_1 + \Delta_1$ are shown in the lower part of Fig. 4 versus T. The dotted-

TABLE IV. Spin-orbit splittings (in meV) at different points of the BZ in gray tin.

Δ_1	Δ_0'	Δ_1'
	Experiment	
475(30) ^a	•	
550 ^b		
400°		
520 ^d	350 ^d	280 ^d
482(6) ^e	300(30) ^e	228(40)°
	Theory	
450 ^f	480 ^f	210 ^f
	540 ^g	4
480 ^h	430 ^h	190 ^h
660 ⁱ	700 ⁱ	370 ⁱ
480 ^j	580 ^j	290 ^j

^aReference 10.

^gG. G. Wepfer, T. C. Collins, and R. N. Euwema, Phys. Rev. B 4, 1296 (1971).

dashed lines correspond to the best linear fit, and the solidlines correspond to an equation similar to Eq. (4) where Θ has been forced to have the same values as for the fit to the corresponding CP energies. The parameters of the fits are listed in Table II. Our values at 100 K, $\Gamma(E_1)=45\pm4$ meV and $\Gamma(E_1+\Delta_1)=52\pm7$ meV, are comparable to those found for Ge,³⁰ 41±3 and 55 ± 7 meV. As in many other zinc-blende materials,⁴⁴ the $E_1+\Delta_1$ structure is slightly broader than the E_1 structure.

These structures are known to be strongly affected by excitonic effects, i.e., electron-hole Coulomb interaction. These effects can be described as the interference of a discrete two-dimensional exciton with a quasicontinuous background. A qualitative description can also be effected by multiplying the one-electron ϵ by a phase factor, as in Eqs. (1) and (2). The phase angle ϕ , plotted in Fig. 5 for the $E_1+\Delta_1$ CP, decreases rapidly with increasing T, indicating a decrease of excitonic effects as T is raised. The value of $\phi=98^\circ$ found at 100 K formally corresponds to a slight mixture of a 2D maximum with a saddle point. This angle increases through the sequence Si, Ge, α -Sn, InSb. 1,30,37

The amplitudes of 2D CP's can be calculated within the one-electron model of Ref. 47:

$$A_{E_1} = 44 \frac{E_1 + \Delta_1/3}{a_0 E_1^2} , (9a)$$

$$A_{E_1+\Delta_1} = 44 \frac{E_1 + 2\Delta_1/3}{a_0(E_1 + \Delta_1)^2}$$
, (9b)

^bReference 13 (from R).

^cReference 13 (from ϵ).

dReference 14.

eThis work.

^fReference 3.

hReference 21.

Reference 25.

Reference 4.

where a_0 is the lattice constant in Å, and the energies are in eV. Using the lattice constant of Ref. 39 and the experimental energies of our present work at 100 K, Eqs. (9) predict $A_{E_1} = 5.6$ and $A_{E_1 + \Delta_1} = 3.4$, with a ratio $A_{E_1}/A_{E_1+\Delta_1}=1.6$. Experimentally, $A_{E_1}=7.1$ $A_{E_1+\Delta_1}=2.5$ are found at 100 K. The agreement is reasonable in view of the crudeness of the theory used. Discrepancies are believed to be due to linear terms in k, 48,49 which increase the transverse mass for the E_1 gap and decrease it for the $E_1 + \Delta_1$ gap, and to an excitonic interaction which increases the amplitude of the E1 structure compared with one-electron calculations.⁵⁰ The first contribution is expected to be small in the case of gray tin because of its large SO splitting Δ_1 ; the latter seems to dominate, as confirmed by the fact that the ratio $A_{E_1}/A_{E_1+\Delta_1}$ decreases from a value of 2.8 at 100 K to 2 at 300 K, where excitonic effects have diminished.

The solid lines of Fig. 3(a) correspond to the best fit to Eq. (3).³⁶ The parameters of the fit are listed in Table I. β should be close to the Debye temperature (Θ_D) . Thus, β is found to decrease through the sequence Ge, α -Sn, InSb, 1,30 following the values of Θ_D . 39 An equally good fit can also be obtained with Eq. (4), which takes into account average statistical factors for absorption and emission of phonons. A similar expression has been used for the fundamental gaps of diamond, silicon, and germanium.⁵¹ Best fits are shown in Fig. 3(a) as dashed lines; fit parameters are listed in Table I. Both fits are equally good; the differences begin at temperatures lower than our experimental ones. Measurements at liquid-He temperatures should decide which of the models yields the best description. The dotted-dashed lines represent a linear fit, and the linear coefficients (Table I) agree within error with those of Ref. 14.

C. E'_0 transitions

 E_0' transitions are believed to take place at the Γ point, or in the Δ direction near the Γ point^{3,4,43} between the $\Gamma_{25'}$ valence bands and the Γ_{15} conduction bands. Four structures are expected for these transitions: $E_0'(\Gamma_8 - \Gamma_6)$, $E_0' + \Delta_0'(\Gamma_8 - \Gamma_8)$, $E_0' + \Delta_0(\Gamma_7 - \Gamma_6)$, and $E_0' + \Delta_0'(\Gamma_7 - \Gamma_8)$, with Δ_0 (Δ_0') the SO splitting of the valence (conduction) band at Γ . The corresponding critical points have a 3D M_0 character. The $E_0' + \Delta_0$ transitions are symmetry forbidden.

The dielectric function for a 3D CP can be approximately described by⁴⁷

$$\epsilon_2 = \frac{2}{3\omega^2} (2\mu^{3/2}\Pi^2)(\omega - E_g)^{1/2} = A(\omega - E_g)^{1/2},$$
 (10)

where μ is the reduced effective mass of the transitions and Π is the corresponding matrix element of \vec{p} . The effective masses of the valence $\Gamma_{25'}$ bands and the conduction Γ_{15} bands can be obtained from 25

$$\frac{2m^*E}{\hbar} = \begin{cases} ak^2 \pm \left[b^2k^4 + c^2(k_x^2k_y^2 + k_y^2k_z^2 + k_z^2k_x^2)\right]^{1/2}, & (11a) \\ a'k^2 - \Delta, & (11b) \end{cases}$$

where $\Delta = \Delta_0$ for the valence bands and Δ'_0 for the conduction bands. The parameters a, b, c^2 , and a' can be calculated from the experimental energy gaps and the matrix elements of \vec{p} between Γ_{15} and Γ_{1} , $\Gamma_{25'}$ for the conduction band, and $\Gamma_{25'}$ and $\Gamma_{2'}$, Γ_{15} , $\Gamma_{12'}$ for the valence band. Using the matrix elements of Ref. 3, we calculate, for the strength A of the CP's, $A_{E'_0} = 3.57$,

$$A_{E'_0 + \Delta'_0} = 1.94$$
, $A_{E_0 + \Delta_0} = 0$, and $A_{E'_0 + \Delta_0 + \Delta'_0} = 0.3$.

We have been able to resolve the E_0' and $E_0' + \Delta_0'$ CP's; the $E_0' + \Delta_0 + \Delta_0'$ CP should be very close to E_2 , and thus too weak to be observed. Line-shape analysis with 3D CP's [see Eq. (2)] yields the following amplitudes, $A_{E_0'} = 4.27$ and $A_{E_0' + \Delta_0'} = 3$, in qualitative agreement with the above predictions. Contributions of transitions slightly away from the Γ point could be responsible for the enhancement of the $E_0' + \Delta_0'$ structure. Our CP energies are about 0.2 eV larger for E_0' than in previous work (see Table III). $E_0' + \Delta_0'$ is also 90 meV larger than the electroreflectance result. Our $\Delta_0' = 300 \pm 30$ meV is in good agreement with electroreflectance and smaller than all theoretical values.

We have assumed Δ'_0 to be T independent and have used the value obtained at 100 K as a fixed parameter while E'_0 and $E'_0 + \Delta'_0$ were fitted simultaneously. The best fit at 100 K is seen in Fig. 2. Owing to the large scattering in the data, attempts to fit temperature dependences to Eqs. (3) and (4) give parameters without physical meaning. Therefore only a linear fit was performed [Fig. 3(a)]. The T-linear coefficients are reported in Table I. The average linear coefficient of E'_0 and $E'_0 + \Delta'_0$ is found to be comparable to that of the E_1 transitions, but is a factor of 3 larger than that found from electroreflectance;14 it is also larger than that in the case of Ge.30 The temperature dependence of the broadening parameter is shown in the upper part of Fig. 4 for E'_0 . The results are similar for the $E'_0 + \Delta'_0$ CP. The large scatter in the data allows us to obtain only its linear coefficient (Fig. 4 and Table II), which is somewhat larger than for the E_1 transitions.

D. E_2 transition

The E_2 transition has the contributions of several critical points. It has been associated with X_4 - X_1 transitions, 10 transitions between the upper valence and lower conduction bands in an extended region near U in the XUK plane,^{3,13} and also with a plateau in the transition energy near $(2\pi/a)(0.75, 0.25, 0.25)$. The best fit to this structure was obtained via a mixture of a 2D maximum and a saddle point; this representation also yields the best fit in the case of Ge.³⁰ The best fit for α -Sn can be seen in Fig. 2. In Fig. 5 we have displayed the phase angle ϕ [Eq. (1)] versus T, together with that of Ge. These ϕ -vs-T plots show a similar behavior, decreasing with increasing T, as for the E_1 transitions. This could indicate that excitonic effects are also important for E2, although different temperature shifts of electronic states contributing to this transition cannot be ruled out as the cause of this dependence. The CP energies and broadening parameters versus T are shown in Figs. 3(b) and 4. The lines

been resolved.

represent the best fits with Eqs. (3)—(5). The fit parameters are listed in Tables I and II. For the energy shift they are similar to those of the lower transitions. The effect of T on the broadening is comparable to that of E_0' , but larger than that for the E_1 transitions, as in the case of Ge.³⁰ The CP energies at 200 K compare well with previous experimental and theoretical work (see Table III).

E. E'_1 transitions

The next structure above the E_2 CP is believed to correspond to transitions between the L_3 valence- and L_3' conduction-band doublets. Four structures, labeled $E_1' + \Delta_1'(L_{4,5}^+ \to L_{4,5}^-),$ $E_1'(L_{4,5}^+ \to L_6^-),$ $E_1' + \Delta_1(L_6^+)$ $\rightarrow L_6^{-1}$), and $E'_1 + \Delta'_1(L_6^+ \rightarrow L_{4,5}^-)$, should be observed; $\Delta_1(\Delta'_1)$ is the SO splitting of the $L_3(L'_3)$ band. If these transitions take place at the L point, the E'_1 and $E'_1 + \Delta'_1 + \Delta_1$ structures should be the larger ones, because they are determined by p_{\perp} matrix elements, while the $E'_1 + \Delta'_1$ and the $E'_1 + \Delta_1$ structures are determined by $p_{\bar{z}}$ matrix elements (the \overline{z} are the {111} directions, and \bot means perpendicular to \overline{z}). In the case of Ge (see Fig. 15 of Ref. 54), the former should be an order of magnitude larger than the latter. Full band-structure calculations³ show, however, two doublets in the (4-5)-eV region: the intensity of the first doublet is almost twice as large as the second. This fact is in agreement with our experiments, although we only have been able to resolve the first doublet. Owing to its energy separation, we have assigned it to the E'_1 and $E'_1 + \Delta'_1$ transitions. A possible explanation for the discrepancy between experiment and the simple two-band model can be the effect of the k-linear coupling on these transitions, as discussed for GaAs in Ref. 49. Critical points could appear in directions perpendicular to the [111] symmetry axis close to the L point (Fig. 1 of Ref. 49) (also see the discussion in Sec. IV F).

We have analyzed these structures with 2D line shapes. The SO splitting $\Delta_1' = 228 \pm 40$ meV found at 100 K was kept constant for the rest of the fits. The CP energies and broadening parameters are shown versus T in Figs. 3(b) and 4. The broadening parameter of the $E_1' + \Delta_1'$ structure is, within experimental error, equal to that of the E_1' structure. The linear coefficients of the CP energies are slightly larger than those of the lower transitions (see Table I). That of Γ is, within experimental accuracy, comparable to those of the other transitions. The value of Δ_1' quoted in Table IV should be taken with care due to the fact that these transitions could correspond to points lying away from the $\{111\}$ directions.

F. Other transitions

Two more structures are still seen in Fig. 2. The small one above the $E_0' + \Delta_0'$ transition at 2.94 eV may correspond to transitions between the second valence band and the lowest conduction band in the [100] direction. Therefore we have labeled it $\Delta_7^{\nu} - \Delta_6^{c}$. It has been theoretically predicted.⁴ However, it has not seen experimentally before. We have been able to resolve it up to 180 K. We have fitted it together with the $E_0' + \Delta_0'$ structure, keeping the parameters obtained before for this CP fixed. Its T dependence is shown in Fig. 3(a). The parameters of the

linear fit, comparable to those of E_2 , are listed in Table I. We have labeled the highest transition seen in Fig. 2, at about 5.2 eV, E_3 , as it corresponds, within 0.2 eV, to the theoretically predicted E_3 structure (transitions in the ΓXUK plane between the upper valence and the third conduction bands). We did not assign it to the $E_1' + \Delta_1' + \Delta_1$ structure because it would lead to a Δ_1 value too large compared to that obtained from the E_1 and $E_1 + \Delta_1$ CP's. However, if the k-linear terms are important, this last possibility should not be disregarded, due to the increase of apparent SO splitting in directions perpendicular to {111}. Its linear coefficient, larger than that for the other structures but comparable within error, is listed in Table I. The E_3 energies are plotted in Fig. 3(b) versus T. This is also the first time that this structure has

G. White tin

The optical constants of evaporated white-tin films have been measured by near-normal-incidence reflectance⁵⁵ in the energy region from 2.1 to 14.5 eV, and the refraction index from 14.5 to 20.5 eV, by the critical-angle method.⁵⁵ Several authors have also measured the optical constants in the infrared region.^{56–58} Band-structure calculations are also available in the literature.⁵⁹

In Fig. 6 we presented the optical constants of the tin film measured at RT after the phase transition took place at $\sim 90\,^{\circ}$ C; this transition temperature agrees well with Raman studies. The phase transition could be monitored by measuring ϵ at the E_2 energy; it produced a large change at this energy (see Figs. 1 and 6). The ϵ 's stabilized about 15 min after the transition. We thus assumed that the transformation was complete. The sample was taken to RT and the ϵ spectra were measured. The optical constants of the β -Sn film may be influenced by the fact that microcracks, due to the increased density of β -Sn, and a rough surface, are formed after the transformation. However, a reasonably good agreement is found with the data of Ref. 55. We have fitted our results with a Drude expression, 42

$$\epsilon(\omega) = 1 - \frac{\omega_p^2}{\omega(\omega + i/\tau)} \ . \tag{12}$$

We obtained, from the fit, $\omega_p = 12 \pm 0.1$ eV and $\tau = (4.1 \pm 0.5) \times 10^{-15}$ s. The fitted curves are plotted as dashed lines in Fig. 6. The deviations between the fit and experimental data are due to interband transitions⁵⁵ at about 1.2 eV, unfortunately outside our energy range. Nevertheless, the it is quite good. The energy-independent relaxation time compares well with the value $\tau = 4.2 \times 10^{-15}$ s of Ref. 55. The plasma frequency is high compared with the value, 8.16 eV, obtained from optical data in the far-infrared region (0-0.1 eV). An effective number of electrons, $n_{\text{eff}} = 2.82$, is obtained from ω_p , instead of 1.33 in Ref. 55. Our value agrees with $n_{\text{eff}} = 2.74$ found for α -Sn in the same energy range.

After a period of about 24 h, the β -Sn film again transforms partially into α -Sn. Once the α - β transition has taken place, β -Sn seems to prevent the full transformation into α -Sn at RT. Thus the transformation is only partly reversible.

V. CONCLUSIONS

We have measured the effect of temperature on optical interband transitions of gray tin between 1.2 and 5.6 eV. Special care was taken with sample preparation and to avoid surface contamination; this allowed us to obtain reliable optical constants for this material. Real and imaginary parts of the dielectric function are presented in the range $100-350~\rm K$. Line-shape analysis of numerically calculated second-derivative spectra of the original data yields information about critical-point energies, broadening parameters, amplitudes, and phases of the E_1 ,

 E_0' , E_2 , E_1' , and E_3 structures, and their corresponding SO-split counterparts. Our results compare reasonably well with available band-structure calculations. To free-electron-like optical constants of the white-tin film obtained after the phase transition induced by heating are well described by a Drude model.

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